

# 1 **Structural Disorder of Graphite and Implications for Graphite** 2 **Thermometry**

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## 10 **Key Points:**

11 graphite, disorder, thermometry, Raman

## 12 **Abstract**

13 Graphitization, or the progressive maturation of carbonaceous material, is considered an irreversible process. Thus, the  
14 degree of graphite crystallinity, or its structural order, has been calibrated as an indicator of the peak metamorphic  
15 temperatures experienced by the host rocks. However, discrepancies between temperatures indicated by graphite crystallinity  
16 versus other thermometers have been documented in deformed rocks. To examine the possibility of mechanical  
17 modifications of graphite structure and the potential impacts on graphite 'thermometry' we performed laboratory  
18 deformation experiments. We sheared highly crystalline graphite powder at normal stresses of 5 and 25 MPa and aseismic  
19 velocities of 1  $\mu\text{m/s}$ , 10  $\mu\text{m/s}$  and 100  $\mu\text{m/s}$ . The degree of structural order both in the starting and resulting materials was  
20 analyzed by Raman microspectroscopy. Our results demonstrate structural disorder of graphite, manifested as changes in the  
21 Raman spectra. Microstructural observations show that brittle processes caused the documented mechanical modifications of  
22 the aggregate graphite crystallinity. We conclude that the calibrated graphite 'thermometer' is ambiguous in active tectonic  
23 settings.

## 24 **1. Introduction**

25 Organic matter, preserved in sedimentary rocks, can be transformed into crystalline graphite due to structural and  
26 compositional changes during diagenesis and metamorphism, a process known as graphitization (Bonijoly et al., 1982;  
27 Wopenka and Pasteris, 1993; Beyssac et al., 2002a; Buseck and Beyssac, 2014; etc.). Graphitization is thought to be an  
28 irreversible process and graphite is known to remain stable to the highest temperatures of granulite facies and the highest  
29 pressures of coesite-eclogite facies (Buseck and Beyssac, 2014). It is generally accepted that the degree of graphite  
30 crystallinity, or its structural order, is determined mainly by the maximum temperature conditions experienced by the host  
31 rocks, whereas lithostatic pressure and shear strain are considered to have only minor influence on graphitization (Bonijoly  
32 et al., 1982; Wopenka and Pasteris, 1993; Bustin et al, 1995). Therefore, graphite crystallinity has been calibrated as an  
33 indicator of the peak temperatures reached during progressive metamorphism (Beyssac et al., 2002a; Reitmeijer and  
34 McKinnon, 1985). However, in strained rocks discrepancies between temperatures indicated by the crystallinity of graphite  
35 vs. other thermometers have been reported (Barzoi, 2015; Nakamura et al., 2015; Kirilova et al., 2017). Thus, numerous  
36 authors have speculated that tectonic deformation results in graphite structural modifications that challenge the validity of  
37 the existing graphite thermometers (Large et al., 1994; Bustin et al, 1995; Crespo et al., 2006; Barzoi, 2015; Nakamura et al.,  
38 2015).

39 Furthermore, graphite occurrence and enrichment have been documented in several fault zones in the world, e. g. the Alpine  
40 Fault zone, New Zealand (Kirilova, et al., 2017), the Hidaka metamorphic belt, Hokkaido, Japan (Nakamura et al., 2015), the  
41 Atotsugawa fault system, Japan (Oohashi, et al., 2012), the Tanakura Tectonic Line, Japan (Oohashi et al., 2011), the Err  
42 nappe detachment fault, Switzerland (Manatschal, 1999), and the KTB borehole, Germany (Zulauf et al., 1990). In these  
43 intensely deformed rocks its presence is of particular interest because its low friction coefficient of  $\mu \sim 0.1$  (Morrow et al.,  
44 2000) allows graphite to act as a natural solid lubricant (Savage, 1948). The mechanical behavior of graphite has been  
45 broadly investigated in both natural and experimental specimens, where it manifests with the lowest  $\mu$  among sheet structure  
46 minerals (Moore and Lockner, 2004; Oohashi et al., 2011, 2013; Rutter, et al., 2013; Kuo et al., 2014, etc.) confirming it  
47 could have a significant impact on fault mechanics. It has been experimentally proven that even a small fraction of graphite  
48 can have a disproportionally large effect on frictional strength where graphite is concentrated by smearing into interlinked  
49 layers (Rutter, et al., 2013).

50 However, structural changes in crystalline graphite caused by tectonic deformation have not yet been systematically  
51 explored. To examine this aspect and to investigate the potential impacts of structural disordering of graphite on the graphite  
52 ‘thermometer’, we have carried out laboratory deformation experiments on highly crystalline graphite powder.

## 53 2. Experimental methods

### 54 2.1 Sample description

55 As a starting material in the current study we used synthetic (commercially synthesized) graphitic carbon to avoid  
56 complexities arising from variable degree of crystallinity in natural carbon materials. Initially, the material was crushed to  
57 average grain size of 100  $\mu\text{m}$  in a RockLabs Swing (TEMA) mill. The resulting fine graphitic powder was annealed at  
58 700°C for two hours in a Lindberg Blue M Muffle Furnace to achieve full graphitization, which is known to occur at this  
59 temperature in the absence of other variations in physical conditions (Buseck and Beyssac, 2014). This was used as the  
60 starting material for the deformation experiments.

### 61 2.2 Experimental procedure

62 In total, 10 deformation experiments were performed at room temperature and room humidity in the Brittle Rock  
63 deformAtion Versatile Apparatus BRAVA (Collettini et al., 2014), at INGV, Rome. For each experiment two 3-mm thick  
64 layers of synthetic graphite gouges were placed in between three grooved forcing blocks in a double-direct shear  
65 configuration (e.g. Dieterich, 1972). The two side blocks are held stationary, and the central forcing block is driven  
66 downward causing shear to occur within the graphite gouge layers. Normal stress is applied by the horizontal piston in load-  
67 feedback control mode and shear displacement accomplished by the vertical piston in displacement-feedback control mode.  
68 Forces are measured with stainless steel load cells ( $\pm 0.03$  kN) and displacements are measured with LVDTs ( $\pm 0.1$   $\mu\text{m}$ )  
69 attached to each piston. Experiments have been conducted at normal stresses of 5 MPa or 25 MPa and aseismic sliding  
70 velocities of 1  $\mu\text{m/s}$ , 10  $\mu\text{m/s}$  and 100  $\mu\text{m/s}$ . The experiments were carried out to total displacements of 20 mm. In addition,  
71 some experiments were stopped at 5 mm and 10 mm and the specimens were then recovered to reveal graphite structural  
72 changes that took place during different amounts of total deformation. The coefficient of friction ( $\mu$ ) was calculated as the  
73 ratio of measured shear load to measured normal load ( $\mu = \tau / \sigma_n$ , where  $\tau$  is shear stress and  $\sigma_n$  is effective normal stress).  
74 The average shear strain within the layer was calculated by dividing shear displacement increments by the measured layer  
75 thickness and summing up. The displacement values of the vertical and horizontal load points were corrected for the elastic  
76 stretch of each load frame, taking into account that the machine stiffness is 1283 kN/mm on the horizontal axis and 928.5  
77 kN/mm on the vertical axis. In addition, we calculated total frictional work for each experiment as a function of shear stress  
78 integrated over the total displacement (Beeler, 2007).

### 79 2.3 Raman microspectroscopy

80 Raman spectra of graphite was measured by an Alpha 300R+ confocal Raman microscope (WITec, Ulm, Germany) with a  
81 532 nm laser (Coherent, Santa Clara, California), located at the Department of Chemistry, University of Otago, New

82 Zealand. The laser (3.0 mW) was focused on the samples with a 50× Zeiss objective. The scattered light was dispersed with  
83 a 1200 g/mm grating. The combination of the 50× objective and 532 nm laser wavelength produced a laser spot size of  
84 approximately 412 nm in diameter. The integration time of each spectrum was 2 seconds with 50 co-additions (100 seconds  
85 in total). The spectra were calibrated using the Raman band from a silicon wafer prior to each set of measurements.

86 The collected spectra were pre-processed in GRAMS AI 9.1 (Thermo Fisher Scientific Inc.), where cosmic spikes were  
87 removed, and a multi-point linear baseline offset was performed. This was followed by peak fitting three Lorentzian-  
88 Gaussian functions to each spectrum with a linear baseline over 1000 - 1700  $\text{cm}^{-1}$ . For each spectrum, the area ratio was  
89 calculated ( $R2 = A_{D1} / (A_G + A_{D1} + A_{D2})$ , where  $A_i$  is the area of the  $i$ th peak, G band is the main high frequency band of  
90 graphite, D1 and D2 bands are defect bands observed in the first order Raman spectrum of graphite) (Wopenka and Pasteris,  
91 1993; Beyssac et al., 2002a).

## 92 **2.4 Scanning electron microscopy**

93 Microstructural analyses of the graphite gouge recovered from the biaxial apparatus were carried out using a scanning  
94 electron microscope (SEM). Some SEM images were acquired from the shiny surfaces of the graphite layers that had been  
95 parallel to the center and or side forcing blocks (Y-Z sections), with a Zeiss Sigma field emission scanning electron  
96 microscope (VP FEG SEM) at the Otago Centre for Electron Microscopy (OCEM), University of Otago, New Zealand. The  
97 instrument was operated in variable pressure mode (VP) at 15 kV using a working distance (WD) of 7 – 8 mm and a VPSE  
98 (VP-mode secondary electrons) detector. In addition, polished thin sections cut perpendicular to the surface of contact with  
99 the center and side forcing blocks (X-Z sections) were imaged on a JEOL JSM-6510 SEM at the University of Potsdam,  
100 Germany, where high-resolution secondary electron images were collected at 20 kV and a WD of 10 mm.

## 101 **2.5 Transmission electron microscopy**

102 Transmission electron microscopy (TEM) was used for detailed microstructural characterization of the shiny surfaces. High-  
103 resolution TEM images were collected by using a JEM-2010 electron microscope, located at the University of Hokkaido,  
104 Sapporo, Japan. The instrument was operated at 200 kV with LaB6 filament. TEM foils (with size of 12 x 5  $\mu\text{m}$  and  
105 thickness of 1  $\mu\text{m}$ ) milled by FIB perpendicular to the shiny surface (X-Z section) were placed on a carbon coated film, and  
106 examined by using dual-axis tilting holder.

## 107 **3. Results**

### 108 **3.1 Mechanical data**

109 Our experiments allowed us to investigate graphite mechanical behavior and structural modifications under various sliding  
110 velocities, normal stresses and shear strain. These conditions are summarized in Table 1. The estimated total frictional work  
111 for each experiment is shown in Table 2.

#### 112 **3.1.1 Friction variations**

113 Over several mm of displacement, the friction coefficient shows a similar evolution trend in all experiments. On a plot of  
114 friction coefficient vs. displacement (Fig. 1a), the friction coefficient ( $\mu$ ) delineates a curve characterized by a rapid increase  
115 to an initial peak friction coefficient ( $\mu_{\text{peak}}$ ), followed by a subsequent exponential decay towards a steady-state friction  
116 coefficient ( $\mu_{\text{ss}}$ ) over a slip weakening distance. The shapes of the friction-displacement curves vary with the normal stress  
117 applied and are steeper for the experiments conducted at 25 MPa than the ones at 5 MPa (Fig. 1a) i.e. the displacement  
118 required to achieve steady-state decreases at higher normal stress. In addition, the values of both  $\mu_{\text{peak}}$  and  $\mu_{\text{ss}}$  (Fig. 1a; Table  
119 1) are significantly lower in the experiments at 25 MPa ( $\mu_{\text{peak}} = \sim 0.4$ ;  $\mu_{\text{ss}} = \sim 0.1$ ) than in the experiments at 5 MPa ( $\mu_{\text{peak}} =$   
120  $\sim 0.5$ ;  $\mu_{\text{ss}} = \sim 0.2$ ) (where  $\mu_{\text{ss}}$  values were read at the end of each experiment). Plots of  $\mu$  at all sliding velocities (Fig. 1a)  
121 show subtle variations in  $\mu_{\text{peak}}$  and  $\mu_{\text{ss}}$  with change of the applied sliding velocities (Fig. 1a; Table 1).

#### 122 **3.1.2 Shear strain variations**

123 Plots of friction coefficient vs. shear strain (Fig. 1b) show significant variations in shear strain attained over equivalent  
124 sliding displacements. The estimated shear strain values are a geometric consequence of different thickness changes, which  
125 are visualized on plots of layer thickness vs. displacement (Fig. 1c). Consideration of the shear strain at equivalent sliding  
126 velocities but different normal stresses demonstrates that shear strains achieved during the 5 MPa experiments are  
127 approximately half of those at 25 MPa (Fig. 1b; Table 1). In addition, the experiments at 25 MPa demonstrate a dramatic  
128 increase in shear strain with increasing slip velocity (Fig. 1b; Table 1), whereas at low normal stress we do not observe any  
129 systematic variations associated with changes in sliding velocities (Fig. 1b, c and d).

### 130 **3.2 Raman spectra of graphite**

131 All the experiments resulted in the development of shiny smooth surfaces with gentle slickenlines (macroscopic fine  
132 grooves, parallel to the slip direction as defined by Toy et al., 2017). Raman spectra obtained on the top of these surfaces,  
133 that had accommodated most of the induced deformation, are compared to Raman spectra from the starting material.

134 Raman data from 20 spectra per sample are presented in Supplementary material 1 (S1). Representative spectra for each  
135 sample are illustrated in Fig. 2, which shows spectra displaying the least (left column) and the most (right column)  
136 disordered graphite within a sample (i.e. lowest and highest R2 values respectively). Spectra that were typical of the average  
137 for each sample are also presented (middle column). Experiments 3 and 7 were stopped at only 5 mm displacement and  
138 resulted in extremely fragile deformed surfaces, which were unable to be extracted without them breaking into pieces too  
139 small to obtain spectra from. Thus, Raman spectra was not measured in these experiments.

140 All the acquired spectra show typical G, D1 and D2 bands, respectively at  $\sim 1580\text{ cm}^{-1}$ ,  $\sim 1350\text{ cm}^{-1}$  and  $\sim 1620\text{ cm}^{-1}$  (S1).  
141 Thus, we could calculate the area ratio R2 for each spectrum (Fig. 2; S1). Raman spectra collected from the starting material  
142 show R2 values ranging from 0 to 0.327 (Fig. 2).. Spectra acquired from the deformed surfaces show higher R2 values (Fig.  
143 2; S1). The most ordered graphite with R2=0.330 was collected in Exp. 2 (Fig. 2) while the most disordered graphite with  
144 R2=0.661 resulted from Exp. 10 (Fig. 2).

145 As R2 values vary within a sample (Fig. 2; S1), we examine average R2 values for each one and compare them with applied  
146 normal stress, sliding velocity, shear strain, and total frictional work (Table 2). The starting material has average  $R2_{\text{pre-shear}}$   
147  $\text{graphite} = 0.173$ , whereas all deformed samples have higher average R2 values (Table 2). Analyzing the average R2 values for  
148 deformed samples reveals that graphite is more disordered in the high normal stress experiments (Table 2) than in the  
149 experiments at 5 MPa. Furthermore, in the experiments at 25 MPa the average R2 increases with increasing sliding velocities  
150 (Table 2). In contrast, at low normal stress, we do not observe any dependence of the degree of graphite structural order on  
151 the applied sliding velocities (Table 2). Overall graphite appears as most disordered in the experiments where the highest  
152 shear strain was achieved (Table 2). The relationship between average R2 and shear strain is illustrated in Fig. 3a by fitting a  
153 power function with a correlation coefficient  $R^2 = 0.95$ . Fitting a power function to average R2 and total frictional work did  
154 not show a consistent correlation (Fig. 3b). The experiments 2 and 6 at low normal stress, which were stopped at 10 mm  
155 displacement and accommodated the least amount of shear strain, contain the least disordered graphite (Fig. 3; Table 2).

### 156 **3.3 Microstructural characteristics**

#### 157 **3.3.1 Scanning electron microscopy (SEM)**

158 Similar microstructural features were observed in all the deformed samples. SEM images obtained from the sample  
159 deformed during experiment 8 are presented to demonstrate our observations (Fig. 4).

160 These high-resolution images in Y-Z sections reveal that the shiny surfaces are decorated by closely spaced (from  $< 5$  to 10  
161 micrometers) slickenlines (Fig. 4a), on top of a smooth continuous layer. In places, the continuity of this layer is interrupted  
162 by fine ( $\sim 1$  to 2 micrometers in width) fractures (Fig. 4a), with random orientation compared to the slip direction.

163 Occasionally, the deformed surface appears as completely disrupted, and is decorated with small graphite grains (from 50 to  
164 <10 micrometers in size), oriented nearly parallel to the shear direction (Fig. 4b). In X-Z sections this highly deformed  
165 surface is observed as a thin slip-localized zone, composed of well-compacted layer of aligned graphite grains (Fig. 4c). This  
166 localized shear surface is underlain by a zone of randomly oriented, inequigranular, irregular graphite grains (Fig. 4d). In  
167 places, most of the graphite grains are aligned with their basal (001) planes parallel to the slip direction, and form compacted  
168 layers, defining a weakly-developed fabric (Fig. 4e). There has been some dilation along these cleavage planes, and the  
169 spaces thus created are filled with smaller graphite grains with their (001) planes sub-perpendicular to the shear direction  
170 (Fig. 4e). Locally, intensely fractured grains are also observed (Fig. 4f).

### 171 **3.3.2 Transmission electron microscopy (TEM)**

172 TEM was used to examine the microstructure of the material that makes up the shiny surfaces (Fig. 4c). TEM analyses were  
173 performed on foils cut perpendicular to this surface. Fig. 5 shows characteristic TEM images obtained from the sample  
174 recovered from experiment 8.

175 Graphite grains in this well-compacted layer have basal planes predominantly aligned with the shear plane, as were observed  
176 in SEM images. However, adjacent grains show slightly different orientations (Fig. 5a). In addition, kink folded graphite  
177 grains are observed in multiple locations in the foils (Fig. 5b, c), which yields a ‘wavy layering’ at a small angle to the shear  
178 direction (Fig. 5b). In isolated areas, there are also some smaller grain fragments with random orientation (Fig. 5d).

## 179 **4. Discussion**

### 180 **4.1 Mechanical behavior**

181 Graphite in our experiments shows mechanical behavior consistent with other mechanical studies of pure graphite gouges.  
182 Our results display low  $\mu_{ss}$  values (from ~0.1 to ~0.2; Table 1) as did the low-pressure deformation experiments of  
183 carbonaceous material performed by Morrow et al. (2000), Moore and Lockner (2004), Oohashi et al. (2011, 2013), Kuo et  
184 al. (2014), and Rutter et al. (2013). The low frictional strength of graphite is well known and has been attributed to its sheet  
185 structure composed of covalently bonded carbon atoms held together only by van der Waals forces. These weak interlayer  
186 bonds along (001) planes are easily broken during shear (Moore and Lockner, 2004; Rutter, et al., 2013). Initial  $\mu_{peak}$   
187 followed by strain weakening during deformation experiments of graphite gouges has been previously explained with the  
188 work involved in rotating the grains with their (001) planes sub-parallel to the shear surfaces, which puts them in the optimal  
189 position for shearing along the weak interlayer bonds (Morrow et al., 2000; Moore and Lockner, 2004; Rutter, et al., 2013).

190 Plots of layer thickness vs. displacement (Fig. 1c) show that the initial compaction resulted in layers with different thickness  
191 prior to shearing. Therefore, we refer to the estimated shear strains (or the ratio of shear displacement to measured layer

192 thickness) as apparent shear strains. Correlation between these values and the conditions of the experiments show that the  
193 apparent shear strains are significantly higher in the experiments performed at 25 MPa than the ones at 5 MPa mainly due to  
194 better compaction of the sheared graphite gouges (Fig. 1c). The apparent shear strains increase with increase in the applied  
195 sliding velocities in the high normal stress experiments (Fig. 1b), however this trend also reflects differences in the layer  
196 thickness prior to shearing (Fig. 1c). The latter interpretation is also supported by the absence of similar correlation during  
197 the low normal stress experiments. Nevertheless, there are too few of these relationships to fully characterize the effect of  
198 normal stress and sliding velocity on shear strain accumulation in graphite gouges, and more mechanical data of this sort  
199 need to be collected in future.

## 200 **4.2 Structural disorder of graphite**

201 We evaluate the structural order of graphite by analyzing variations in R2 ratios, which depend on the increase of defect  
202 bands (D1 and D2) in the Raman spectrum of graphite. Previous studies have documented increases in R2 with decreasing  
203 degree of graphite crystallinity (Wopenka and Pasteris, 1993; Beyssac et al., 2002a, b). Therefore, our experimental study  
204 may demonstrate transformation of fully/highly crystalline graphite (with R2 ratios ranging from 0 to 0.327; Fig. 2; S1) into  
205 comparatively poorly organized graphitic carbon (with R2 ratios up to 0.661; Fig. 2; S1). However, we do not have direct  
206 evidence suggesting intragranular deformation. Alternatively, the observed increase of defect bands is also likely to reflect  
207 increase in the grain boundary density (Tunistra and Koenig, 1970; Pimenta et al., 2007). The latter is further supported by  
208 our microstructural data that reveals the existence of fine graphite grains (ranging from 50  $\mu\text{m}$  to nm scale grains; Fig. 4, 5)  
209 which are significantly smaller in size than the average grain size of the starting material (100  $\mu\text{m}$ ). We interpret that shear  
210 deformation caused intense grain size reduction, and hence increase in the grain boundary density, that was reflected in the  
211 Raman spectra as a decrease of the overall aggregate crystallinity.

212 We also acknowledge that the slickenlined surfaces that were produced experimentally contain some graphite that yield  
213 spectra comparable to those acquired from the starting material i.e. there is highly ordered graphite that appears as unaffected  
214 by the deformation. However, at least some of these spectra are derived from undeformed graphite powder that underlies the  
215 shear surfaces and could not be entirely removed during sample preparation due to the fragile nature of the samples. It is also  
216 possible that some non-deformed graphite powder was accidentally measured through the fractures that are cross-cutting the  
217 accumulated shear surfaces (Fig. 4a). But even if some graphite did not undergo mechanical modification during the  
218 experiments, the results overall validate that structural disorder of graphite aggregates can result from shear deformation  
219 subsequent to the graphitization process.

220 To understand the potential causes for the documented structural disorder of graphite, we compared the measured average  
221 R2 with the parameters of the performed experiments. Our data show a good correlation between the average R2 and the  
222 apparent shear strain at the tested aseismic sliding velocities (Fig. 3). However, these bulk shear strains (Table 1) are likely



223 to be significantly lower than the shear strains accommodated within the thin shear surface and thus, we refer to the above  
224 relationship as a rough approximation. Nevertheless, previous authors have also suspected that shear strain may play an  
225 important role for graphite modifications, and evidence for this has been found in graphite crystallinity variations in natural  
226 samples from active fault zones (Kirilova et al, 2017; Nakamura et al., 2015), and strained rocks in metamorphic terrains  
227 (Barzoi, 2015; Large et al., 1994). Thus, we speculate that shear strain may play an important role for the final structural  
228 order of graphite aggregates and consequently, the previously proposed model of progressive graphitization due to increase  
229 of temperature (Bonijoly et al., 1982) does not completely reflect the graphite formation mechanisms.

230 Our microstructural observations provide some indications of the deformation processes that affected graphite structural  
231 order. The shiny slickenlined surfaces are composed of very fine-grained material visible as slip-localized zone on SEM  
232 images (Fig. 4d). Nanoscale observations reveal graphite grains within it occasionally form stacked kink-band structures,  
233 (Fig. 5b, c). This zone, which we assume accommodated most of the induced deformation, is underlined by a less deformed  
234 zone composed of larger graphite grains in a finer matrix that in places has developed as an anastomosing fabric, typical of  
235 creeping gouges (Fig. 4d). In rare places at SEM scale brittely fractured grains also occur (Fig. 4f and 5d). The interpreted  
236 structures suggest that brittle processes operated during shearing, and we conclude that these processes resulted in the  
237 structural disorder of graphite, manifested as changes in the Raman spectra. This interpretation is in agreement with the  
238 conditions of our experiments (i.e. shearing with aseismic velocities took place at room temperature conditions), that  
239 typically would not induce temperatures high enough for ductile processes. Furthermore, the microstructures and the inferred  
240 processes are exactly the same as those observed by Nakamura et al. (2015) in the Hidaka metamorphic belt, Japan.

241 However, crustal fault zones do not only accommodate brittle deformation. At higher temperatures and confining pressures,  
242 localised shearing can operate by crystal plastic mechanisms (White et al., 1980). We hypothesize that graphite structural  
243 order could be influenced by ductile deformation, as was also suggested in previous studies by Large et al. (1994), Bustin et  
244 al. (1995), Barzoi et al. (2015). Furthermore, Kuo et al. (2014) and Oohashi et al. (2011) simulated fault motions in synthetic  
245 and natural carbonaceous material with variable degree of maturity at the start of the experiments (ranging from amorphous  
246 carbonaceous material to crystalline graphite). Both studies reported graphitization of carbonaceous material due to localized  
247 frictional heating rather than structural disordering. These experiments reveal the impact of seismic velocities on graphite  
248 structural order and the fact their findings differ so markedly from ours further highlights the complexity of graphite  
249 transformations in fault zones.

### 250 **4.3 Implications for graphite thermometry**

251 The structural order of graphite measured by Raman spectroscopy has been applied as a thermometer that relies on  
252 progressive maturation of originally-organic carbonaceous material during diagenesis and metamorphism. Previous studies  
253 have focused on calibrating this thermometer. The current best calibration is described by the following equation  $T (^{\circ}\text{C}) = -$

254 445 \* R2 + 641 ± 50 (Beysac et al. 2002) by inferring a linear correlation between R2 ratio and peak metamorphic  
255 temperatures. However, this thermometer disregards the effects of mechanical modifications on the structure of graphite  
256 aggregates, which this study has identified as having a substantial influence on the R2 ratios in deformed graphite gouges at  
257 sub-seismic velocities.

258 Our experiments demonstrate an increase of the R2 ratio of initially highly crystalline graphite powder due to brittle  
259 deformation (Fig. 3a; Table 2). In natural analogues, the pre-shear graphite would yield temperatures up to 641 ± 50 °C (S1),  
260 which is the upper limit of the calibrated thermometer (Beysac et al. 2002). Whereas, the sheared samples would indicate  
261 peak metamorphic temperatures as low as 347 ± 50 °C (S1). Thus, we experimentally prove that in active tectonic settings  
262 graphite thermometers may underestimate the peak metamorphic temperatures by < 300 °C. In cataclasites from the Alpine  
263 Fault zone, New Zealand (Kirilova et al., 2017) and fault zones of the Hidaka metamorphic belt, Japan (Nakamura, et al.,  
264 2015), the graphite thermometer yields temperature discrepancies of more than 100 °C compared to temperature estimates  
265 derived both from the surrounding high-grade amphibolite facies mylonites and the lower grade equilibrium cataclastic  
266 phases (marked by chlorite alteration). Barzoi (2015) also described differences of ~ 150 °C in graphite temperatures  
267 between strained and less strained low grade metamorphic rocks from Parang Mountains, South Carpathians.

268 We conclude that the existing graphite thermometer is unreliable in active tectonic settings. Furthermore, a calibration of this  
269 thermometer may be impossible to achieve because both structural disorder of graphite and graphitization (Oohashi et al.,  
270 2013) are likely to be encountered in fault zones.

## 271 **5. Conclusions**

272 We have experimentally demonstrated graphite structural disorder, manifested as changes in the Raman spectra, by  
273 performing shear deformation experiments at aseismic sliding velocities insufficient to generate appreciable frictional heat  
274 on graphite gouges composed of powdered highly-organized graphite. Microstructural data presented here reveal that this is  
275 a result of brittle processes. Our findings clearly compromise the validity of the calibrated graphite thermometer by showing  
276 it may significantly underestimate the peak metamorphic temperatures in active tectonic settings.

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285 experimental procedures.

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354

355 **Table 1.** Summary of the conditions at which experiments were carried out and results.

356 **Table 2.** Summary of the relationship between shear strain and average R2 within a sample. The conditions of each  
357 experiment are also given as follows: applied normal stress in MPa, sliding velocities in  $\mu\text{m/s}$  and sliding displacement in  
358 mm.

359 **Figure 1.** Plots of mechanical data (a) friction coefficient,  $\mu$  vs. displacement (b) friction coefficient,  $\mu$  vs. shear strain (c)  
360 layer thickness vs displacement.

361 **Figure 2.** Representative Raman spectra illustrating: (i) the most structurally ordered graphite (left column) within a sample;  
362 (ii) graphite with average structural order per sample (middle column); and (iii) the most disordered graphite (right column)  
363 encountered in each sample. The R2 ratio for each spectrum is also noted in italic font.

364 **Figure 3.** Plot of the average R2 ratio per sample vs (a) the shear strain accumulated during each experiment (b) the total  
365 frictional work estimated for each experiment.

366 **Figure 4.** SEM images, obtained from the deformed graphite gouge during experiment 8 (normal stress at 25 MPa with 1  
367  $\mu\text{m/s}$  sliding velocity), show: (a) Slickensides ornamenting the shear surface; (b), (c) A well-compacted layer of aligned  
368 graphite grains, which make up the shear surface. Bright patches due to a differential charging effect; (d) A less deformed  
369 zone with typical cataclastic fabric, underlying the shear surface; (e) Dilated cleavage planes in large graphite grains filled  
370 with smaller platy graphite grains oriented sub-perpendicular to the shear direction; (f) Fractured graphite grains.

371 **Supplementary material 1 (S1).** Raman data from 20 spectra per sample together with calculated R2 ratio and average R2  
372 value for each sample. The last column represents temperature estimated by the current best calibration of a Raman-based  
373 thermometer:  $T (^{\circ}\text{C}) = - 445 * R2 + 641 \pm 50$ .

374

Experiment number	Normal stress (MPa)	Sliding velocity ( $\mu\text{m/s}$ )	Displacement (mm)	Peak friction coefficient ( $\mu_{\text{peak}}$ )	Steady state friction coefficient ( $\mu_{\text{ss}}$ )	Shear strain maximum
1	5	1	20	0.53	0.22	17.70
2	5	1	10	0.53	0.22	8.17
3	5	1	5	0.52	<i>not reached</i>	4.23
4	5	10	20	0.53	0.24	20.45
5	5	100	20	0.57	0.22	16.89
6	5	100	10	0.55	0.22	9.80
7	5	100	5	0.57	<i>not reached</i>	3.87
8	25	1	20	0.43	0.17	21.45
9	25	10	20	0.43	0.17	31.86
10	25	100	20	0.41	0.14	46.77

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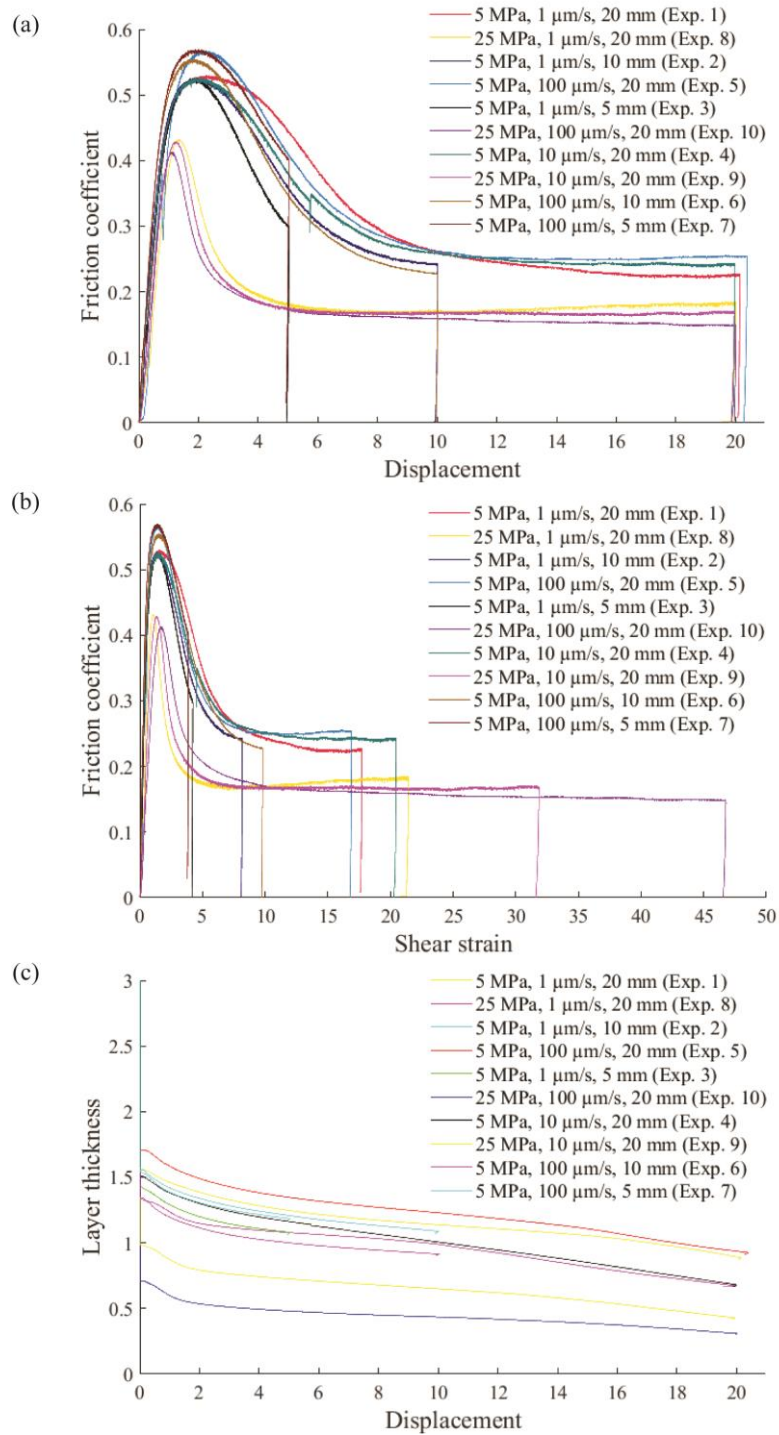
**Table 1.** Summary of the conditions at which experiments were carried out and results.

Sample	Experimental conditions	Shear strain	Average R2 (error estimate $\pm 0.05$ )	Total frictional work
<b>Pre-shear graphite</b>	N/A	N/A	0.173	
<b>Exp. 2</b>	5 MPa, 1 $\mu\text{m/s}$ , 10 mm	8.17	0.438	17.711
<b>Exp. 6</b>	5 MPa, 100 $\mu\text{m/s}$ , 10 mm	9.80	0.430	17.7506
<b>Exp. 5</b>	5 MPa, 100 $\mu\text{m/s}$ , 20 mm	16.89	0.454	32.0578
<b>Exp. 1</b>	5 MPa, 1 $\mu\text{m/s}$ , 20 mm	17.70	0.506	25.885
<b>Exp. 4</b>	5 MPa, 10 $\mu\text{m/s}$ , 20 mm	20.45	0.517	30.5943
<b>Exp. 8</b>	25 MPa, 1 $\mu\text{m/s}$ , 20 mm	21.45	0.520	96.6089
<b>Exp. 9</b>	25 MPa, 10 $\mu\text{m/s}$ , 20 mm	31.86	0.580	92.4834
<b>Exp. 10</b>	25 MPa, 100 $\mu\text{m/s}$ , 20 mm	46.77	0.604	87.703

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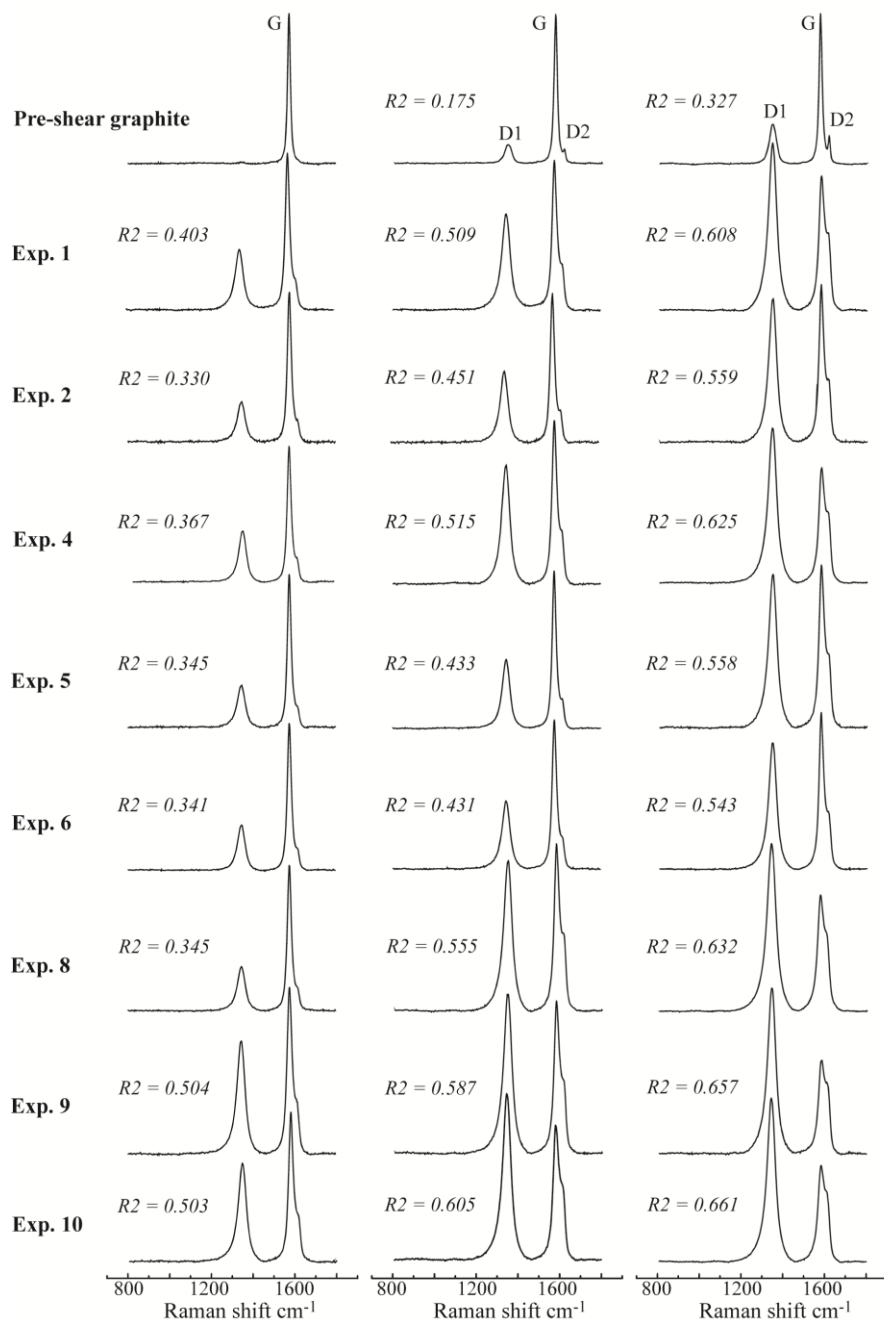
379 **Table 2.** Summary of the relationship between shear strain, average R2, and total frictional work within a sample. The  
380 conditions of each experiment are also given as follows: applied normal stress in MPa, sliding velocities in  $\mu\text{m/s}$  and sliding  
381 displacement in mm.





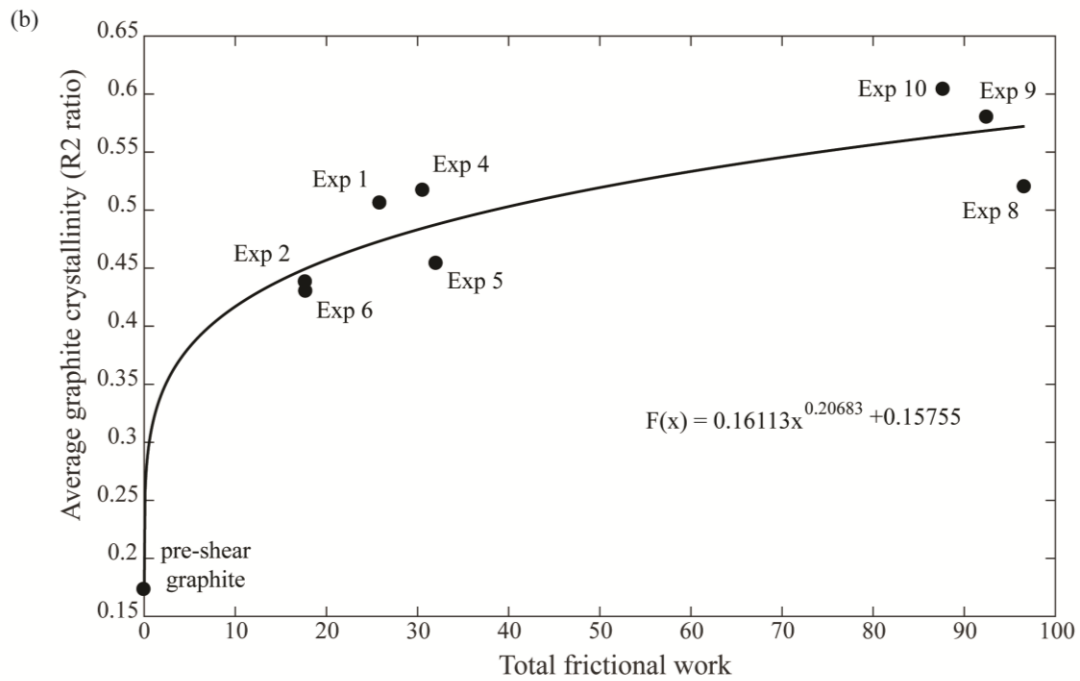
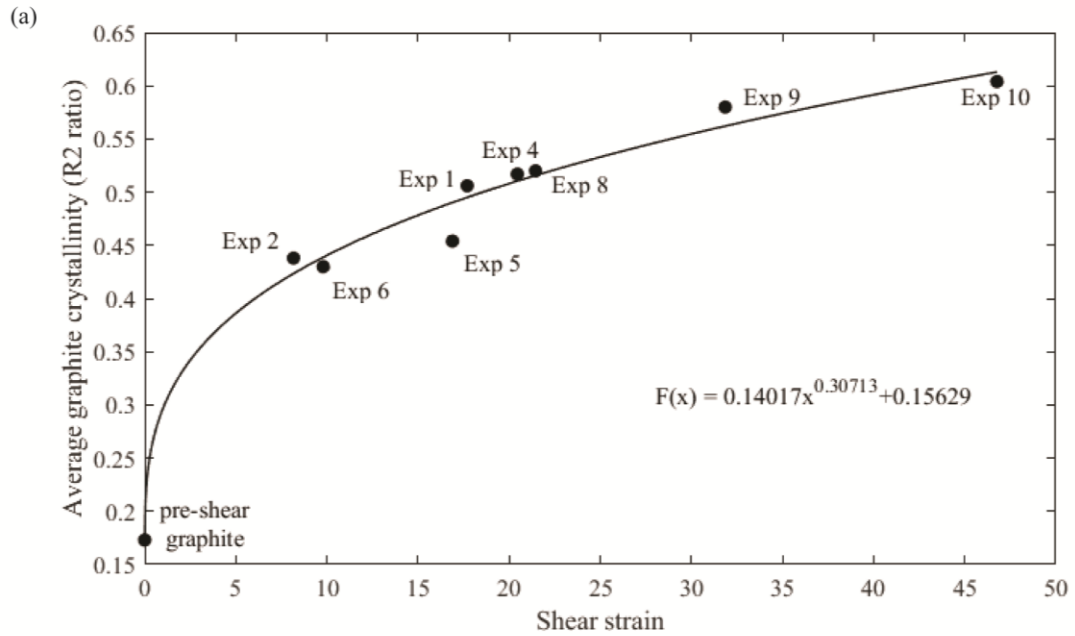
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383 **Figure 1.** Plots of mechanical data (a) friction coefficient,  $\mu$  vs. displacement (b) friction coefficient,  $\mu$  vs. shear strain (c)  
 384 layer thickness vs. displacement



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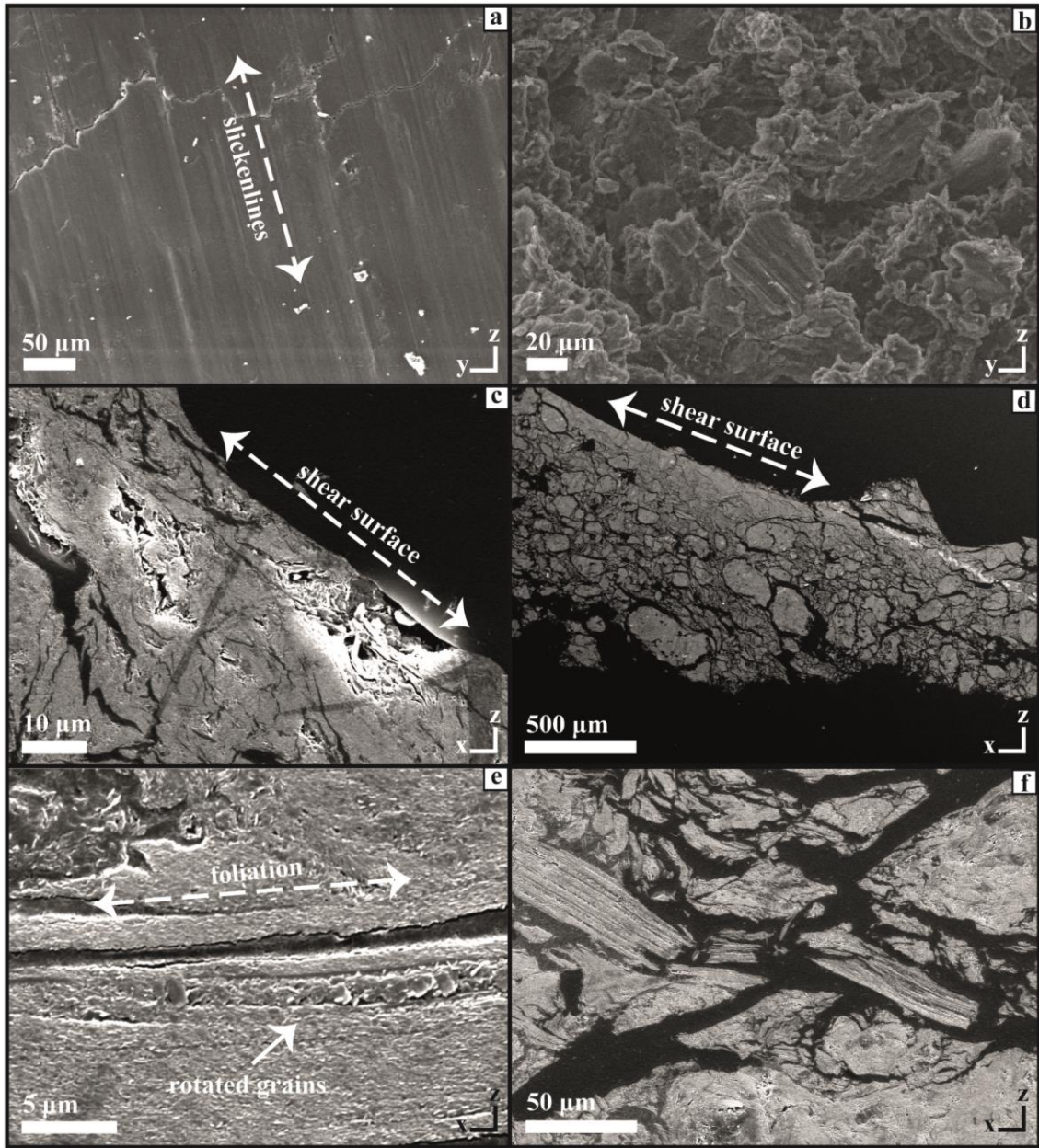
387 **Figure 2.** Representative Raman spectra illustrating: (i) the most crystalline graphite (left column) within a sample; (ii)  
 388 graphite with average crystallinity per sample (middle column); and (iii) the most disordered graphite (right column)  
 389 encountered in each sample. The  $R_2$  ratio (with an error estimate of 0.05) for each spectrum is also noted in italic font.



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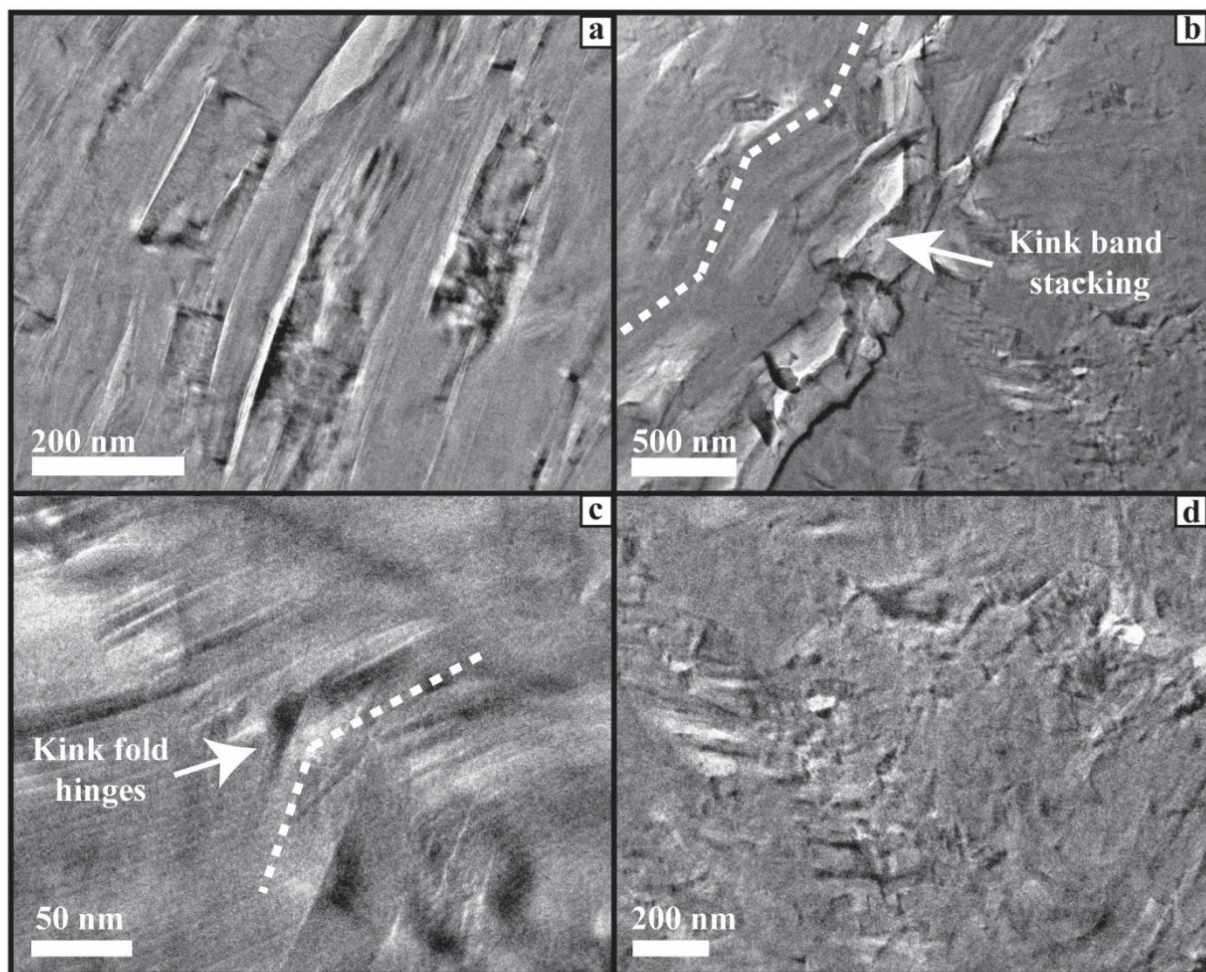
391 **Figure 3.** (a) Plot of the average R2 ratio vs. shear strain accumulated during each experiment. (b) Plot of the average R2  
 392 ratio vs. total frictional work during each experiment.

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395 **Figure 4.** SEM images, obtained from the deformed graphite gouge during experiment 8 (normal stress at 25 MPa with 1  
 396  $\mu\text{m/s}$  sliding velocity), show: (a) Slickenlines ornamenting the shear surface; (b), (c) A well-compacted layer of aligned  
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 398 zone with typical cataclastic fabric, underlying the shear surface; (e) Dilated cleavage planes in large graphite grains filled  
 399 with smaller platy graphite grains oriented sub-perpendicular to the shear direction; (f) Fractured graphite grains.



**Figure 5** TEM images showing microstructural characteristics of the slip-localized shear surface: (a) aligned grains showing slightly different orientation; (b) kink band stacking; (c) dilated kink fold hinges; (d) fragmented grains.